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L6

(FILE 'HOME' ENTERED AT 17:18:16 ON 22 SEP 2001)

FILE 'REGISTRY' ENTERED AT 17:18:20 ON 22 SEP 2001

FILE 'REGISTRY' ENTERED AT 17:18:30 ON 22 SEP 2001

17 (2<AL<20 AND 0<ZN<10 AND 0<SN<15 AND 0<MN<2 AND 50<MG)/MAC

FILE 'HCA' ENTERED AT 17:19:14 ON 22 SEP 2001

10 L1

SELECT L2 IPC 1 3 7

13 96440 E1-3

9321 (MAGNESIUM OR MG) AND (TIN OR SN) AND (ZINC OR ZN) AND (ALUMINU 1189 L3 AND L4

129 L5 AND GRAIN?

71:41699 HCA AN

Extruded and formed objects from magnesium-lithium ΤI

Foerster, George S. IN

Dow Chemical Co. PA

SO Fr., 4 pp. CODEN: FRXXAK

DTPatent

LΑ French

FAN.CNT 1

PATENT NO. KIND DATE APPLICATION NO. DATE ------ 19680802 FR 19670621 _____

PΙ FR 1534886

The addn. of certain quantities of Pb or \mathbf{Sn} to $\mathbf{Mg}\text{-Li}$ AB alloys, followed by extrusion and hardening by annealing, increases the alloys' resistance. In particular the apparent limit of elasticity during tension, the resistance to rupture during tension, and the corrosion resistance in saline water are all increased. The process consists of heating to 300-425.degree., an Mg alloy contg. Li (0.4-2.0), \mathbf{sn} (3-20) or Pb (3-20) or a mixt. of Pb and \mathbf{sn} (total 3-20), \mathbf{Al} (0-4), \mathbf{Mn} (0-2), \mathbf{Zn} , (0-3), and \mathbf{Zr} (0-0.2%). The hot alloy is extruded under pressure through a die at a speed of at least 15 m./min. and is then hardened by annealing at 120-320.degree. immediately after extrusion. Alternatively alloys contg. s_n can be advantageously quenched in water immediately after extrusion and then annealed as above. At 175.degree. satisfactory hardening occurs within 24 hrs.

Example's Copy

68:89493 HCA AN

Powder-metallurigical manufacture of magnesium alloys ΤI

Foerster, George S. IN

PA Dow Chemical Co.

SO Ger., 4 pp. CODEN: GWXXAW

DTPatent

LΑ German

FAN.CNT 1

PATENT NO. KIND DATE APPLICATION NO. DATE ____ _____ DE 1259578 19680125 PΙ 19590501 PRAI US

In the manuf. of Mg alloys having a heterogeneous and AΒ finely grained structure because of a finely dispersed metal phase in the alloy, molten Mg or a molten Mg alloy contg. metal addns., e.g., Mg9Ba, Mg2Co, Mg2Ge, Mg3Sb2, Mg2Si, or Mg9Sr which are sol. in the melt, but sol. only up to 0.1% below the solidus temp., are sprayed in the form of small droplets, cooled below the solidus temp., and mech. worked in a known manner, e.g., by extrusion pressing. The mech. strength of the Mg alloys may be further improved by addns. of Mn .ltoreq.2.5, Al .ltoreq.13, Zn .ltoreq.9, Ag .ltoreq.16, Bi .ltoreq.11, Ca .ltoreq.1, Li .ltoreq.15, sn .ltoreq.16, Zr .ltoreq.1, Th .ltoreq.8, rare earth metals .ltoreq.2%.

> <13 Al £9 3n 616 Sn 52.5 Mm 42

Eraman's Cops

71:104643 HCA AN

TΙ Magnesium alloys

IN Winter, Heinrich

Norsk Hydro-Elektrisk Kvaelstofaktieselskab PA

SO Ger., 2 pp. CODEN: GWXXAW

DTPatent

German LΑ FAN.CNT 1

PATENT NO.	KIND	DATE	APPLICATION NO.	DATE
PI DE 1301914	В	19690828	DE 1967-N30523	19670517
NO 120334	В	19701005	NO 1968-1929	19680516
		40650545		

PRAI DE 1967-N30523 19670517

Mg alloys of high mech. strength at high temp.

(200.degree.) are made by incorporating rare metals, such as Ce, in the $\mathbf{M}\mathbf{g}$ matrix in the form of silicides. The rare metal silicide forms a ternary eutectic with the Mg and is distributed within the

Mg matrix. Addnl. elements, such as Al, Zn,

Mn, Ca, Ag, Cd, Sn, and Be can also be incorporated to further improve the mech. properties. Thus, a Mg. alloy was made from Al 3, z_n 1, Ce 3.25, Si 1.34 wt. %,

Mg balance. This alloy, when kept at 200.degree. for 100 hrs. under a load of 3 kg./mm.2, exhibited an elongation of 0.37%, while the same alloy without Ce showed an elongation of 0.74% under the same conditions. The alloys are suitable for casting, particularly when the following elements are included (max. shown), Zn 7,

Al 10, Mn 2, Ca 1, Ag 5, Cd 5, Sn 5, and Be

0.01 wt. %.

\(\frac{18Ca}{3.252RBM}. \)
 \(\frac{72Bn}{52Sn} \)
 \(\frac{18Ca}{3.252RBM}. \)

Examiner's 6pm

71:24334 HCA AN Heat-resistant magnesium alloy with good casting TI properties Winter, Heinrich IN Norsk Hydro-Elektrisk Kvaelstofaktieselskab PA SO Norw., 4 pp. CODEN: NOXXAJ DTPatent Norwegian LA FAN.CNT 1 APPLICATION NO. DATE PATENT NO. KIND DATE _____ NO 20675 19681117 PΙ 19670517 PRAI DE This Mg-base alloy contg. Al AΒ .ltoreq.10, Zn .ltoreq.7, .ltoreq.Mn .ltoreq.2, Ca .ltoreq.1, Ag .ltoreq.5, Cd .ltoreq.5, Sn .ltoreq.5, Be .ltoreq.0.01 wt. % is characterized by also contg. 0.5-5% rare earth metals, e.g., Ce, and 0.2-3% Si. It has improved tensile strength when hot, e.g., at 200.degree..

Examer's Gry

71:73500 HCA AN Deformed magnesium-base alloy
Drits, M. E.; Sviderskaya, Z. A.; Trokhova, V. F.
Baikov, A. A., Institute of Metallurgy ΤI IN PΑ U.S.S.R., Otkrytiya, Izobret., Prom. Obraztsy, Tovarnye Znaki 1969, SO 46(14), 76 CODEN: URXXAF ${\bf DT}$ Patent Russian LΆ FAN.CNT 1 PATENT NO. KIND DATE APPLICATION NO. DATE _____ _____ 19690418 SU SU 241679 19670929 PΙ AB To increase the plasticity, the following alloy was produced: Li 7-9, Al 4-6, Sn 2-4, Zn 0.8-2, Mn -0.15-0.5, Na .ltoreq.0.01, and Mg the remainder.

the fair

- AN 76:102994 HCA
- TI Structure and properties of semifinished products made of ultralight alloys
- AU Drits, M. E.; Gur'ev, I. I.; Sviderskaya, Z. A.; Elkin, F. M.; Trokhova, V. F.
- CS USSR
- SO Tekhnol. Legk. Splavov. Nauch.-Tekh. Byul. Vses. Inst. Legk. Splavov (1971), No. 2, 9-16
 From: Ref. Zh., Met. 1971, Abstr. No. 10I546
- DT Journal
- LA Russian
- Data are given on mech. and phys. properties of pressed and rolled Mg-Li alloys contg. Li 5, Zn 1, Sn 1, Mn 0.4, rest Mg (IMV1); Li 8, Al 5, .degree.n 1, Cd 4, Mn 0.4, rest Mg (IMV2); Li 8, Al 5, Zn 1, Sn 3, Mn 0.4, rest Mg (IMV 2-1); and Li 14, Zn 5, Sn 0.2, and Mg traces (IMV3). IMV2 has the most favorable combination of strength and plasticity: Brinell hardness 79, tensile strength (.sigma.B) 28.3, yield point (.sigma.T) 21.8 kg/mm2, and elongation .delta. 6.8 in the longitudinal direction and 27.4, 20.5, and 9.7, resp., in lateral direction at a much higher compressive strength than std. Mg alloys. Mech. properties and stress-rupture strength, detd. during a 75-hr test at 600-100.degree., indicate that IMV1, IMV2, and IMV2-1 alloys can be used up to 100.degree. and IMV3 alloy up to 60.degree.

Gaminer's Ggs

AN 77:8981 HCA

- TI Ultralight magnesium-lithium alloys based on the .alpha.+.beta. two-phase region
- AU Drits, M. E.; Sviderskaya, Z. A.; Trokhova, V. F.
- CS USSR
- SO Strukt. Svoistva Legk. Splavov (1971), 118-22. Editor(s): Korol'kov, A. M. Publisher: "Nauka", Moscow, USSR. CODEN: 24YAAA
- DT Conference
- LA Russian
- A study of the effect of alloying addns. on the structure and the AΒ properties of Mg alloys with Li belonging to the two-phase .alpha. + .beta. region (7-10 wt. % Li) established the pos. effect of Al, Sn, Cd, Zn, and Mn on the properties of these alloys. The best combination of strength properties and plasticity in the deformed state was achieved for alloys contg.: Li 7.0-9.0, Al 4.0-6.0, Sh 2.0-4.0, Zn 0.8-2.0, Mn 0.15-0.5%, rest Mg; and Li 7.0-10.0, Al 4.0-6.0, Cd 3.0-5.0, Zn 0.8-2.0, Mn 0.15-0.5%, rest Mg. Both alloys can easily be extruded on heating the rods to 280-320.degree. and can be rolled into sheets at 300-50.degree.. Mech. properties of the alloys prepd. by testing of hot-extruded rods and of sheet material on fabrication of the samples in longitudinal and transverse directions relative to the rolling direction, were detd. respect to strength and plasticity, the alloys are not as good as those of .alpha.-phase Mg-Li alloys, but they are considerably better than those based on the .beta.-phase. For sheet material, there is practically no anisotropy in the longitudinal and the transverse directions. A study of the microstructure of the alloys and of the effect of individual alloying components shows that the main factor providing high strength is the presence of a Mg-rich solid soln. of a complex compn. with not only Al and Li addns., but also Sn and (or) Cd addns. playing a strengthening role. The presence of Li also increases the plasticity.

Examiner's Cops

AN 82:159590 HCA

TI Magnesium alloy

IN Al'tman, M. B.; Malinkovich, A. N.; Mitrofanova, M. F.; Blyablin, A. A.; Shmilovich, A. R.

PA USSR

SO U.S.S.R.

From: Otkrytiya, Izobret., Prom. Obraztsy, Tovarnye Znaki 1974, 51(39),

CODEN: URXXAF

DT Patent

LA Russian

FAN.CNT 1

PATENT NO.	KIND	DATE	APPLICATION NO.	DATE
SU 447452	T	19741025	SU 1968-1500810	19681213

PI SU 447452 T 19741025 SU 1968-1500810 19681213

AB Sn is added to the Mg alloy to stabilize its mech. properties. Thus, the Mg alloy contained Li 7.9, Al 3.5-4.5, Zn 1-2, Mn 0.3-0.5, misch metal 0.2-0.8, and Sn 0.3-0.8%. The misch metal contained Nd 55-60, La 25-30, and Ce 8-10%.

3.5-4.5 Al 1-2 Zn 0.3-0.8 Sn 0.3-0.5 Mn Examiner's Gpg

AN 114:28384 HCA

TI Manufacture of vibration-damping magnesium alloy by ingot casting

IN Yamauchi, Goro; Mino, Masato

PA Nippon Telegraph and Telephone Corp., Japan

SO Jpn. Kokai Tokkyo Koho, 3 pp. CODEN: JKXXAF

DT Patent

LA Japanese

FAN.CNT 1

PATENT NO. KIND DATE APPLICATION NO. DATE

JP 02047238 A2 19900216 JP 1988-197520 19880808

PI JP 02047238 A2 19900216 JP 1988-197520 19880808

AB The vibration-damping Mg alloys for structural parts are prepd. by inoculating molten Mg with H and 0.1-10%

Al, Si, P, Ca, Ti, V, Cr, Mn, Fe, Co, Ni, Cu, Zn, Ga, Ge, Y, Zr, Nb, Mo, As, Cd, In, Sn, Sb, and/or Bi as a solid soln. or hydride. The inoculated Mg-alloy melt is cast, and the resulting ingots are forged, rolled, and then heat-treated in flowing H at 90.degree.-solidus temp. The alloy product shows a good vibration damping at .gtoreq.1 kHz.